Synthesis and characterization of LaCoO₃ based cathode and its chemical compatibility with CeO₂ based electrolytes for intermediate temperature solid oxide fuel cell (ITSOFC)

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Received 27 February 2006; revised received 30 October 2006; accepted 19 January 2007

The use of alternate cathode materials with improved performance and without any chemical reaction with adjoining electrolyte is required for a reduction in operating temperature of SOFC from 1273 K to about 1073 K (ITSOFC). Cobalt containing perovskite oxides such as LaCoO₃ tend to exhibit a higher ionic conductivity due to a greater concentration of oxygen vacancies than other perovskite oxides. The mixed ionic and electronic conducting cathode of the La_{1-x}Sr_xCoO_{3-δ} systems has shown the lowest cathodic overpotential for an SOFC air electrode. In this research work, fine powders of La_{0.70}Sr_{0.30}CoO_{3-δ} (LSC) cathode and Ce_{0.90}Gd_{0.10}O_{2-δ} (GDC) and Ce_{0.80}Sm_{0.20}O_{2-δ} (SDC) are synthesized by glycine nitrate combustion synthesis and systematically characterized by XRD and particle characteristics. The electrical properties of LSC cathode and GDC and SDC electrolytes are also studied. But, the crucial requirement for applicability of LSC cathode is its chemical compatibility in conjunction with the alternate solid electrolytes, GDC and SDC without any phase formation. The XRD studies showed no reaction products when the La_{0.70}Sr_{0.30}CoO_{3-δ} cathode is mixed and calcined with GDC and SDC electrolyte at 1573 K. Hence, the LSC cathode may be combined with CeO₂ based electrolytes in ITSOFC application.

Keywords: Solid oxide fuel cell, Perovskite oxide, Intermediate temperature

IPC Codes: C22B59/00, H01M8/00

Solid oxide fuel cells (SOFCs) have been attracting a great attention as a promising technique for electrical power generation because of potential for high-energy conversion efficiency (>70%) and low environmental pollution. In general, they must be operated at high temperature between 1173 to 1273 K because the Y_2O_3 stabilized ZrO₂ (YSZ) is used as the usual electrolyte material and has low ionic conductivity at operating temperatures lower than 1073 K. Such high operating temperatures of SOFC lead to some serious problems, such as, physical and chemical degradation of components. Therefore, it is desirable to develop SOFCs operating at intermediate temperatures below 1073 K. To reduce the operating temperature, two approaches are under active considerations as follows: one is development of suitable alternative to Sr doped LaMnO₃ (LSM) cathode and another is development of promising alternative to Y₂O₃ stabilized ZrO₂ (YSZ) electrolyte¹⁻³.

Doped lanthanum cobaltite $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ is of great interest as potential cathode material for

ITSOFCs⁴⁻⁵. Also, it is reported that doped ceria electrolytes such as $Ce_{0.90}Gd_{0.10}O_{2-\delta}$ and $Ce_{0.80}Sm_{0.20}O_{2-\delta}$ have shown greater oxide ion Scm^{-1}) conductivities (>0.1 in intermediate temperature range⁶⁻⁷. For using LSC ceramic as cathode in ITSOFC, it's interface stability with GDC/SDC electrolytes became a major concern. In this research work, LSC cathode and GDC/SDC electrolyte materials by glycine-nitrate combustion method in the form of fine powder have been synthesized. The physico-chemical properties of these powders are systematically evaluated. The chemical compatibility between the above cathode and ceria based electrolytes is also studied. The obtained results are discussed in order to qualify them as alternative components for ITSOFC application.

Experimental Procedure

Preparation of powders

 $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ cathode and $Ce_{0.90}Gd_{0.10}O_{2-\delta}$ / $Ce_{0.80}Sm_{0.20}O_{2-\delta}$ electrolyte powders were synthesized

method^{8,9}. by glycine-nitrate combustion Stoichiometric amounts of corresponding metal nitrates were dissolved in distilled water along with appropriate quantity of glycine (fuel and complexant) in a quartz crucible. The resulting clear solution was introduced into a preheated muffle furnace maintained at 823 K. The mixture then froths, undergoes dehydration, ignites and burns to yield voluminous and foamy homogeneous fine ceramic powders. Thus, combustion reaction was completed with in a few seconds. When glycine is the fuel the following reaction (Eq. 1) takes place. 1 mole of glycine gives 5 moles of gases.

 $2 C_2 NH_5 O_2 + 9/2 O_2 \longrightarrow N_2(g) + 4CO_2(g) + 5H_2 O(g) \dots (1)$

The stoichiometric redox reaction between metal nitrates and glycine to produce substituted $LaCoO_3/CeO_2$ can be represented by the following theoretical equations (Eqs 2, 3 & 4):

 $0.70La(NO_3)_3 + 0.30Sr(NO_3)_2 + Co(NO_3)_2 + 2.612NH_2CH_2COOH + 0.252O_2 \longrightarrow La_{0.70}Sr_{0.30}CoO_{2.85}(s) + 5.224CO_2(g) + 3.6561N_2(g) + 6.53H_2O(g)$ (15.4101 mol of gases/mol La_{0.70}Sr_{0.30}CoO₃₋₈) ... (2) $0.90 Ce(NO_3)_2 + 0.10 Gd(NO_3)_2 + 1.67$

$$NH_{2}CH_{2}COOH + 0.2325 O_{2} \longrightarrow Ce_{0.90}Gd_{0.10}O_{1.95}(s) + 3.34 CO_{2}(g) + 2.335 N_{2}(g) + 4.175 H_{2}O(g)$$
(9.85 mol of gases/mol Ce_{0.90}Gd_{0.10}O_{2-\delta}) ... (3)

Calcination of as-synthesized powders was carried out in alumina crucibles at 1073 K for 6 h in air to remove deposited carbon and unreacted organic residues and to get phase pure compounds¹⁰.

Results and Discussion

Powder XRD studies

Sr doped LaCoO3

The $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ cathode powder is crystallized to a single-phase perovskite (rhombohedralhexagonal system) with no intermediate phases such as La₂O₃, La₂CO₃ and CO₃O₄ which have been observed in other synthesis processes like acetate or nitrate processes¹¹. The Fig. 1 shows the XRD pattern of the La_{0.70}Sr_{0.30}CoO₃₋₈ powder. The unit cell parameters 'a' (= 5.4264) and 'c' (=13.2576) agree well with those reported (a = 5.4270 and c = 13.2180) in the JCPDS card No. 36-1393 (for La_{0.60}Sr_{0.40}CoO₃₋₈).

Doped Ceria

The obtained XRD patterns of both heat-treated $Ce_{0.90}Gd_{0.10}O_{2-\delta}/Ce_{0.80}Sm_{0.20}O_{2-\delta}$ powders had a cubic (f.c.) fluorite crystal structure. The observed XRD patterns of the doped ceria electrolytes are shown in Figs 2 to 3. All peaks in the XRD patterns are very sharp showing the crystalline nature of the heat-treated powders. The XRD patterns of the doped CeO₂ are matched with the data for CeO₂, which is indexed in JCPDS card No. 34-0394. No other peak corresponding to any impurity is observed in the XRD patterns of the doped ceria powders. Not much change



Fig. 1-XRD pattern obtained on La_{0.70} Sr_{0.30}CoO₃₋₈ powder



Fig. 2-XRD pattern obtained on Ce0.90Gd0.10O2-8 powder



Fig. 3-XRD pattern obtained on Ce_{0.80}Sm_{0.20}O₂₋₆ powder

in the lattice parameters (a = 5.39664 for GDC and a = 5.41839 for SDC) is observed.

Powder characteristics

The particle characteristics data obtained on La_{0.70}Sr_{0.30}CoO_{3-δ} and Ce_{0.90}Gd_{0.10}O_{2-δ}/Ce_{0.80}Sm_{0.20}O_{2-δ} electrolyte powders clearly indicate the fine quality of combustion-synthesized materials. The prepared cathode/electrolyte were subjected to particle size measurements using Horiba Laser Particle Size Analyzer (LA-910) with triple distilled water as medium. The obtained particulate properties are indicated in Table 1. The mean particle diameter represents the arithmetically averaged value of the frequency distribution. The BET surface area of the doped based electrolyte materials was measured using the nitrogen gas adsorption multipoint BET method assuming the cross sectional area of $16.2 \times 10^{-20} \text{ m}^2$ for a nitrogen molecule using H₂ as a carrier gas at a temperature of 473 K with relative gas pressure range of 0.3. The bulk/tapped density⁹ for these powders was also measured and reported. The increase in mean particle size of SDC is due to the formation of agglomerates after heat treatment. The particle size, density and surface area appear to depend mainly upon the number of moles of gaseous products liberated (Eqs 1, 2, 3 & 4) during combustion⁸. The escaping gases not only make the combustion residue fine but also help to dissipate the heat, thereby preventing the oxides from sintering and consequently, more porous oxides (with high surface area, lower density and particle size) are obtained.

Electrical conductivity studies

Sr doped LaCoO₃

The d.c. electrical conductivity for LSC cathode was measured on the sintered (at 1673 K for 5 h) rectangular specimens (Length = ~ 2.25 cm,



Fig. 4-Electrical conductivity of $La_{0.70}Sr_{0.30}CoO_{3.\delta}$ pellet measured in air

Table 1—Particulate properties obtained on alternative
cathode / electrolyte powders

Sl. No.	Sample	Mean particle diameter (µm)	BET surface area (m²/g)	Bulk density (g/ce)	Tap density (g/cc)
L	$La_{0.70}Sr_{0.30}CoO_{3-\delta}$	9.25	2.573	0.200	0.417
2	$Ce_{0.90}Gd_{0.10}O_{2.\delta}$ (GDC)	9.794	2.213	0.2160	0,4909
3	$\frac{Ce_{0.80}Sm_{0.20}O_{2.\delta}}{(SDC)}$	61.212	2.432	0.1820	0.4136

Breadth = ~ 0.5 cm and Thickness = ~ 0.5 cm) using a four-probe method between 297 - 1178 K. Fig. 4 shows the electrical conductivity of $La_{0.70}Sr_{0.30}CoO_{3.\delta}$ measured in air as a function of temperature. From the Fig. 4, it is evident that the electrical conductivity of LSC decreases with temperature upto 1178 K (1324 Scm⁻¹). LSC has shown electrical conductivity of 3038 Scm⁻¹ at 297 K (initial temperature). A similar behaviour of $La_{1-x}Sr_{x}CoO_{3-\delta}$ (where $\times \ge 0.3$) is also reported in literature¹². This decrease in conductivity with increasing temperature suggests that the conductivity for LSC specimen is metallic at all temperatures. The electrical conductivity value $(\sigma \approx 10^2 \text{ Scm}^{-1})$ at the operating temperature of SOFC is acceptable for the cathode¹³. From these results, it that LSC exhibits high electrical drawn is conductivity suitable for cathode in SOFC for intermediate temperature operation.

Doped Ceria

The electrical properties of sintered (at 1573 K for 3 h) circular GDC/SDC (diameter = 1 cm) components were studied by A.C. impedance spectroscopy in air. The A.C. impedance spectra of

GDC electrolyte specimen obtained at 773, 873 and 973 K is indicated in Figs 5(a), (b) and (c). From the plots, it is noted that as the temperature is increased, the time constants of the relaxations resulting from the individual polarizations are reduced and the arcs are shifted to higher frequencies as reported earlier¹⁴. The A.C. impedance spectra of SDC electrolyte specimen obtained at 773, 873 and 973 K is indicated in Figs 6(a), (b) and (c). From the impedance plots, the individual resistance R_i of the specimens can be converted to conductivity, σ_i using the Eq. (5)

$$\sigma_i = l / SR_i \qquad \dots \qquad (5)$$

where l is the sample thickness and S the electrode area of the sample surface. The calculated conductivity values are indicated in Table 2. From the Table 2, it is found that the conductivities of both the specimens (GDC/SDC) are in line with the reported data¹⁴.



Fig. 5-(a), (b) & (c) The A.C. impedance spectra of $Ce_{0.90}Gd_{0.10}O_{2\cdot\delta}$ electrolyte specimen obtained at 773, 873 and 973 K

Chemical compatibility studies - Reaction experiments

this research work, La_{0.70}Sr_{0.30}CoO_{3-δ} In is examined as alternative cathode material for ITSOFC requirement application. The crucial for its chemical compatibility applicability is its in conjunction with the alternate solid electrolytes, $Ce_{0.90}Gd_{0.10}O_{2-\delta}$ (GDC) and $Ce_{0.80}Sm_{0.20}O_{2-\delta}$ (SDC) without any new phase formation. Therefore, the

Table 2—Electrical conductivity data obtained on doped ceria electrolyte specimens					
Sample	Temperature (K)	Electrical conductivity (S/cm)			
$\begin{array}{c} Ce_{0.90}Gd_{0.10}O_{2\cdot\delta}\\ (GDC) \end{array}$	773 873 973	0.000051015 0.00035711 0.001190			
Ce _{0.80} Sm _{0.20} O _{2-δ} (SDC)	773 873 973	0.00006422 0.0004816 0.001605			



Fig. 6-(a), (b) & (c) The A.C. impedance spectra of $Ce_{0.80}Sm_{0.20}O_{2\cdot\delta}$ electrolyte specimen obtained at 773, 873 and 973 K

chemical compatibility between the above cathode and GDC and SDC based solid electrolytes is investigated.

The $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ cathode and $Ce_{0.90}Gd_{0.10}O_{2-\delta}$ (GDC) and $Ce_{0.80}Sm_{0.20}O_{2-\delta}$ (SDC) electrolyte powders were mixed individually in a 1:1 weight ratio, together in a pestle and mortar. Each powder (cathode + electrolyte) mixture was ground well, placed in a 20 mm diameter die, compacted with an uniaxial hydraulic pressing machine at a load of 156 Mpa. Each pellet was then annealed in air at 1573 K for 3 h. The temperature of 1573 K was selected for the chemical compatibility tests, because, although the operating temperature of the cell is substantially less, the steps involved in the fabrication of ceramic cathode/ electrolyte/ anode structures usually require processing up to at least 1473 K^{15-16} . The calcined pellets were crushed into powder and the reaction products were identified by X-ray diffraction (XRD). Prior to XRD, the microstructures of annealed samples were investigated by scanning electron microscopy.

XRD studies

LSC + GDC

The comparison of X-ray diffraction data of the individual powder (LSC and GDC) and mixed powder (LSC + GDC) is shown in Table 3. It is reported that La_{0.80}Sr_{0.20}CoO_{3- δ} showed no reaction with GDC, whereas, it reacted with YSZ when annealed at 1473 K¹⁷. From the Table 3, it is noted that La_{0.70}Sr_{0.30}CoO_{3- δ} when calcined with Ce_{0.90}Gd_{0.10}O_{2- δ} at 1573 K for 3 h did not form new reaction products. Also, it is seen that the peaks observed at 2 θ = 33.0°, 47.0°, 58.7°, 68.8° and 78.2° in the XRD pattern of LSC + GDC are assigned to both LSC and GDC

phases, since the d values of both the samples are similar. As there is no new reaction product detected, it is drawn that the chemical compatibility between Sr-doped LaCoO₃ and GDC solid electrolyte is good.

LSC + SDC

The comparison of X-ray diffraction data of the individual powder (LSC and SDC) and mixed powder (LSC + SDC) is shown in Table 4. It is reported that the YSZ electrolyte coated with SDC interlayer was used to protect the solid-state reaction with LSC cathode in $SOFC^{18}$. From the table, it is noted that the $La_{0.70}Sr_{0.30}CoO_{3\cdot\delta}$ calcined with $Ce_{0.80}Sm_{0.20}O_{2\cdot\delta}$ at 1573 K for 3 h showed no reaction products. Also, it is seen that the peaks observed at $20 = 32.8^{\circ}$, 46.9° and 59.0° are indexed to both LSC and SDC phases, since the d values of both the samples are same. As no reaction products could be detected, it is drawn that there is chemical compatibility between $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ cathode and $Ce_{0.80}Sm_{0.20}O_{2-\delta}$ electrolyte.

SEM studies

LSC + GDC

shows the microstructure Figure 7 of а $La_{0.70}Sr_{0.30}CoO_{3-\delta}+Ce_{0.90}Gd_{0.10}O_{2-\delta}$ after pellet sintering at 1573 K for 3 h. It is seen that a well defined boundary between the two phases (LSC/GDC) exists. The GDC grains are found in between LSC grains in some places. The grain size of LSC is 50 µm. The grain size of GDC is 25 µm. The surface of the pellet is smooth.

LSC + SDC

Figure 8 shows the microstructure of a $La_{0.70}Sr_{0.30}CoO_{3-\delta}+Ce_{0.80}Sm_{0.20}O_{2-\delta}$ pellet after sintering at 1573 K for 3 h. It is noted that the surface is completely covered by LSC grains. LSC grains are

Table 3Comparison of XRD data of individual powder with the XRD data of $La_{0.70}Sr_{0.30}CoO_{3.6}$ +
$Ce_{0.90}Gd_{0.10}O_{2-8}$ powder mixture after annealing at 1573 K for 3 h

XRD data for LSC + GDC				XRD data for LSC		XRD data for GDC	
powder mixture							
Peak No.	20 values	d values	Peak assigned for	20 values	d values	20 values	d values
1	28.200	3.162	GDC			28.800	3.097
2	33.000	2.712	LSC/GDC	33.400	2.681	33.300	2.688
3	47.000	1.932	LSC/GDC	47.800	1.901	47.700	1.905
4	55.800	1.646	GDC			56.600	1.625
5	58.700	1.572	LSC/GDC	59.200	1.559	59.300	1.557
6	68.800	1.363	LSC/GDC	69.400	1.353	69.600	1.350
7	76.100	1.250	GDC			76.900	1.239
8	78.200	1.221	LSC/GDC	79.500	1.205	79.300	1.207

$Ce_{0.80}Sm_{0.20}O_{2-\delta}$ powder mixture after annealing at 1573 K for 3 h								
XRD data for LSC + SDC powder mixture			XRD data for LSC		XRD data for SDC 8020			
Peak No.	20 values	d values	Peak assigned for	20 values	d values	20 values	d values	
1	28.100	3.173	SDC			28.600	3.119	
2	32.800	2.728	LSC / SDC	33.400	2.681	33.100	2.704	
3	46.900	1.936	LSC / SDC	47.800	1.901	47.500	1.913	
4	55.700	1.649	SDC			56.300	1.633	
5	59.000	1.564	LSC / SDC	59.200	1.559	59.000	1.564	

Table 4—Comparison of XRD data of individual powder with the XRD data of $La_{0.70}Sr_{0.30}CoO_{3-\delta} + Ce_{0.80}Sm_{0.20}O_{2-\delta}$ powder mixture after annealing at 1573 K for 3 h



Fig. 7-SEM photograph obtained on $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ + $Ce_{0.90}Gd_{0.10}O_{2-\delta}$ pellet after sintering at 1573 K for 3 h



Fig. 8-SEM photograph obtained on $La_{0.70}Sr_{0.30}CoO_{3-\delta}$ + $Ce_{0.80}Sm_{0.20}O_{2-\delta}$ pellet after sintering at 1573 K for 3 h

larger (100 μ m). The SDC grains are seen in between the LSC grains. The grain size of SDC is smaller, (50 μ m). The pore size is 10 μ m. Also, it is seen that there exists well defined boundary between LSC and SDC grains. No new grains are observed.

Conclusion

The glycine nitrate combustion synthesis is a simple method to prepare ultrafine particles of

cathode/electrolytes for SOFC application. The La_{0.70}Sr_{0.30}CoO_{3- δ} cathode showed no reaction products when mixed and calcined with Ce_{0.90}Gd_{0.10}O_{2- δ} and Ce_{0.80}Sm_{0.20}O_{2- δ} electrolyte. Hence, this LSC cathode is useful with doped ceria electrolytes in ITSOFC.

Acknowledgement

One of the authors (ASN) is thankful to the Council of Scientific and Industrial Research (CSIR), Government of India for awarding Senior Research Fellowship to carry out his Ph.D. research work in CECRI. He also thanks the management of Karunya University for its encouragement and interest in this research activity.

References

- Fukui Takeshisa, Ohara Satoshi, Murata Kenji, Yoshida Hiroyuki, Miura Kazuhiro & Inagaki Taru, *J Power Sources*, 106 (2002) 142.
- 2 Fagg D P, Abrantes J C C, Perez-Coll D, Nunez P, Kharton V V & Frade J R, *Eletrochim Acta*, 48 (2003) 1023.
- 3 Dyck C R, Yu Z B & Krstic V D, Solid State Ionics, 171 (2004) 17.
- 4 Kahoul A, Hammouche A, Naamoune F, Chartier P, Poillerat G & Koenig J F, *Mater Res Bull*, 35 (2000) 1955.
- 5 Singh R N & Lal B, Indian J Chem, 40 (2001) 1037.
- 6 Thangadurai V & Weppner W, *Electrochim Acta*, 49 (2004) 1577.
- 7 Song H Z, Wang H B, Zha S W, Peng D K & Meng G Y, Solid State Ionics, 156 (2003) 249.
- 8 Muthuraman M, Arul Dhas N & Patil K C, Bull Mater Sci, 17 (1994) 977.
- 9 Samson Nesaraj A, Studies on Materials and Components for Intermediate Temperature Solid Oxide Fuel Cells, Ph.D. Thesis, Alagappa University, Karaikudi, 2002.
- 10 Hatchell C, Sammes N M & Brown I W M., J Power Sources, 77 (1999) 64.
- 11 Hwang Hae Jin, Moon Jooho, Awano Masanobu & Maeda Kunihiko, *J Am Ceram Soc*, 83 (2000) 2852.
- 12 Takeda Y, Veno H, Imanishi N, Yamamoto O, Sammes N & Phillips M B, *Solid State Ionics*, 86-88 (1996) 1187.

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- 13 Sasaki Y, Takeda Y, Kato A., Imanishi N, Yamamoto O, Hattori M, Iio M & Esaki Y, Solid State lonics, 118 (1999) 187.
- 14 Zhan Zhongliang, Wen Ting-Lian, Tu Hengyong & Lu Zhi-Yi, J Electrochem Soc, 148 (2001) A 427.
- 15 Steele B C H, J Power Sources, 49 (1994) 1.

- 16 Wiik Kjell, Schmidt Christian R, Faaland Sonia, Shamsili Senem, Einarsrud Mari-Ann & Grande Tor. J Am Ceram Soc, 82 (3) (1999) 721.
- 17 Ralph J M, Schoeler A. & Krumpelt M, J Mater Sci, 36 (2001) 1162.
- 18 Uchida Hiroyuki, Arisaka Shin-ich & Watanabe Masahiro, Electrochem Solid State Lett, 2 (1999) 428.